APPLICATION NOTE

Raman, PL & AFM measurements on lithographically-created structures
Raman, PL & AFM measurements on lithographically-created structures

Silicon (Si), being the second most common element in the earth’s crust, is the material most widely used for integrated circuits as well as solar cells. Modern integrated circuit devices work with structures down to 14 nm in size[1] and while the latest developments use a variety of different technologies, traditionally the structures in semiconductor devices are produced using lithography. In order to ensure the highest quality in the devices, the production steps need to minimize stress and stress induced artifacts. Other structures in which layers with different lattice constants are grown on top of each other require the strain for lattice matching. It has been shown [2,3] that Raman measurements can effectively probe the strain states of crystalline materials and provide an effective, nondestructive way of developing the most appropriate production processes. From solar cell research it is additionally known that stress and crystallinity in Si has a strong effect on the photoluminescence (PL) signal of the Si [4].

In this application note we report on structures in crystalline Si created by laser lithography which were in turn examined on the same instrument using confocal Raman imaging, confocal PL imaging and AFM.

Experimental

The experiments were performed using an alpha300 RA confocal Raman-AFM. The XY positioning of the sample for scanning and lithography was achieved using a piezoelectric scanning stage. A stepper motor was used for the focus control (10 nm step size). With this system all experiments could be performed as follows:

a) Lithography
The system was equipped with a pulsed 532 nm laser (typically employed for fluorescence lifetime imaging microscopy – FLIM) which was used for laser writing in combination with the DaVinci Lithography Package. The CW power of the laser was 15 mW with a repetition rate of 20 MHz and a pulse duration of 20 ms. The scripting function of the lithography package allowed not only the arbitrary movement of the sample in XY but also the upward movement of the microscope (to defocus the laser beam) when a movement without laser lithography is desired. Alternatively, an electronic shutter could have been used to achieve this.

b) Confocal Raman Imaging
An additional frequency-doubled Nd:YAG laser emitting 532 nm (CW – continuous wave) was used for confocal Raman imaging at 20 mW. The 100x NA 0.9 objective allowed the laser to focus to a diffraction-limited spot with a diameter of 355 nm. The signal was collected using a 25 μm core diameter optical multimode fiber which acted as a pinhole to achieve the best depth resolution. A UHTS 300 spectrometer in combination with a 1800 g/mm grating and a back-illuminated CCD camera were used for detection. The integration time per spectrum was 12.2 ms. Two scans were performed on the laser lithographed area: A planar scan in XY (50 x 50 μm² with 200 x 200 points) an a depth scan in the XZ plane (20 x 5 μm² with 200 x 50 points).

c) Confocal PL Imaging
For confocal PL imaging the same laser as for Raman imaging was used. In this case a specially selected 50x NA 0.8 objective was chosen which allowed the best throughput for the PL signal (in the range of 1000 - 1250 nm for Si) while showing minimal chromatic aberration between the excitation wavelength (532 nm) and the PL peak. A 100 μm core diameter multimode fiber was used to collect the signal and deliver it to a SpectraPro 2300i mirror-based spectrometer using a 150 g/mm grating and an InGaAs CCD camera 1024 pixels in width. A scan of 50 x 50μm² with 100 x 100 points and 0.21 s integration time per spectrum was performed.

d) Atomic Force Microscopy
The measurements were performed in AC mode using a force-modulation cantilever (Nanoworld) with a resonance frequency of 87 kHz. The first overview scan was performed on an area of 50 x 50 μm with 512 x 512 points. A second scan was performed with 14 x 14 μm² and 256 x 256 points.

Fig. 1: White-light image of the lithographically-created structure in Si.
Results

The lithography process for creating the structure took approximately 2 minutes at the specified laser power. Following this, the white-light image as shown in Figure 1 could be seen. The image was recorded with the WITec Control software using the built-in video camera. Following the planar confocal Raman image scan, Figure 2a could be extracted from the integrated intensity of the first order Si band near 520 rel. cm⁻¹. Figure 2b shows the intensity profile along the blue cross section shown in Figure 2a. The image as well as the cross section show an increased intensity of the Si signal close to the laser written structure as well as a strong decrease exactly along it. The cross section reveals a FWHM of the intensity profile of 680 nm. As the system is diffraction-limited, the laser for writing the structures was as narrow as 355 nm full width at half maximum (FWHM) (0.6 x NA = 0.6 x 532 nm/0.9 = 355 nm). When scanning the same structures again with diffraction-limited resolution the Gaussian profile of the laser during the Raman measurement is convoluted with the geometry of the structure itself. Since the scribing is a thermal process, it is not surprising that the detected structures are slightly larger than diffraction-limited. The brighter and darker areas in Figure 2a suggests that this could be caused by differences in the height of the sample. In such a case the brighter areas could be lying in an optimal focal plane and the dark areas in planes which are not well in focus. In order to check this, a depth scan was performed along the line marked in green in Figure 2a. The result can be seen in Figure 3a where the integrated intensity of the first order Si peak is shown. It can clearly be seen that the intensity originates from approximately the same height and that only very little signal is originating from the positions where the laser-written structures are. It can additionally be noted that brighter areas are located right next to the structures. This is in agreement with the results as seen in Figure 2a where higher intensity is detected right next to the structures. Figure 3b shows the intensity profile extracted along the turquoise line. The sharpness of this profile is partly due to the high depth resolution achievable with the system and partly due to the limited penetration depth of the laser light into Si.

![Fig. 2: Confocal Raman image of the intensity of the first order Si line [a] and the intensity profile along the cross section marked in blue [b]. The green line in [a] indicates where the depth scan was performed.](image)

![Fig. 3: Confocal Raman image of the intensity of the first order Si line along a depth scan [a] and the intensity profile along the cross section marked in turquoise [b].](image)
In order to probe the strain state of the Si around this structure, the data acquired in the planar as well as in the depth scan was analyzed in terms of peak shift of the first order Si peak. For the analysis, the advanced fitting option in the WITec Project Plus software was used and a Lorentzian curve was fitted to the spectra. From the results, the exact position of the peaks could be extracted and this is plotted for the planar scan in Figure 4a and for the depth scan in Figure 4c. Figure 4b shows two representative, averaged spectra with the corresponding fitted Lorentzian curves. The red curve is representative of the red regions and the blue curve of the blue-purple regions. The peak shift can clearly be seen from this and it is also obvious that the Lorentzian fit follows the data points extremely well, which also demonstrates the imaging quality of the UHTS 300 spectrometer (see also [5]). The shift to lower wavenumbers at the edges of the structures is obvious in both scans. This is indicative of compressive stress in the crystalline structure at these positions.

In order to further examine the laser-written scribed structures, a confocal PL scan was performed as well. Figure 5a shows the integrated intensity of the Si PL signal and in Figure 5b the PL spectrum itself is shown. This spectrum was acquired with a 210 ms integration time right in the center of the structure and only the camera background signal was subtracted (no smoothing or averaging was applied).

The second order peak of the excitation wavelength at 1064 nm can also be seen in this spectrum. As Si is an indirect semiconductor, the PL efficiency is extremely low compared to i.e. GaAs. Therefore, only a highly effective detection beam path allows the measurement of the PL signal from a diffraction-limited point with integration times similar to those necessary for Raman imaging. Similar to the previously seen confocal Raman image (Figure 2a), the intensity of the signal is stronger close to the structures. The signal was also examined for peak shifts, but did not show a significant change. It has been shown, however, that more elaborate fitting models also allow the identification of the stress in Si structures from the PL signal [4].

Fig. 4: Confocal Raman image of the position of the first order Si line in the planar scan [a] and in the depth scan [c]. Two representative spectra showing the shift are shown in [b].

Fig. 5: Confocal PL image of the intensity of the PL signal of Si in a planar scan [a] and a representative PL spectrum which originates from the center of the structure [b].
The PL and the Raman scans clearly show that where the laser writing took place, the structures emitted weaker Raman and PL signals. Looking at the depth scan, however, it seemed puzzling that inside the structures no signal was detected. If the laser writing had generated grooves then one would have expected the signal at a lower depth if the grooves were not extremely narrow and deep. In order to clarify this question, AFM scans were performed on the structure. Figure 6a shows an AFM overview scan and Figure 6b a zoomed-in scan focusing on one of the arrows. Both AFM images clearly show higher topography along the structure and from Figure 6b it can clearly be seen that dot-like structures are present along the written structure. At a distance of more than 10 μm from the structure, none of these features were visible. The extent of the elevated structures range from about 1 to 2 μm FWHM and the dots approximately range in diameter from 300-600 nm. The AFM images indicate that significant structural changes occurred in the Si where the laser writing took place. The significantly lowered Si signal near 520 rel cm⁻¹ in combination with the AFM images could indicate that the crystalline structure of the Si was destroyed. In such a case one could expect to see traces of amorphous Si along the structures. In order to check this, the average spectrum far away from the structure and the average spectrum inside the grooves were calculated from the planar Raman data set and are shown in Figure 7.

The spectra clearly show the strong decrease in the intensity of the first order Si peak, but also an increase in the signal background. It is striking that the background is higher on the low-wavenumber side of the Si peak compared to the high wavenumber side of it. Since amorphous Si shows a broad peak in this region, one could postulate that the material generated through the laser writing is in some ways related to amorphous Si.

Fig. 6: AFM scan of the structure. Overview [a] and zoomed-in scan on one of the structures [b].

Fig. 7: Raman spectra well outside the structure (red) and inside the grooves (blue).
Conclusion

In a single microscopy system laser lithography, confocal Raman imaging, confocal PL imaging and AFM were performed on a crystalline Si sample. All experiments were controlled through the same electronics (alphaControl) and the same software (WITec Control). The written structures are clearly detectable with all measurement techniques. The Raman as well as the PL signal originating from crystalline Si is strongly reduced at the laser written structures and a broad enhancement of the signal background of the Raman Si signal can be detected. AFM measurements allowed the exclusion of narrow grooves being the origin of the reduced Raman and PL signal and showed that the structures consist of dot-like elevations.

References